Automation methodologies and large-scale validation for *GW*: Towards high-throughput *GW* calculations

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(Received 26 June 2017; revised manuscript received 25 September 2017; published 17 October 2017)

The search for new materials based on computational screening relies on methods that accurately predict, in an automatic manner, total energy, atomic-scale geometries, and other fundamental characteristics of materials. Many technologically important material properties directly stem from the electronic structure of a material, but the usual workhorse for total energies, namely density-functional theory, is plagued by fundamental shortcomings and errors from approximate exchange-correlation functionals in its prediction of the electronic structure. At variance, the GW method is currently the state-of-the-art *ab initio* approach for accurate electronic structure. It is mostly used to perturbatively correct density-functional theory results, but is, however, computationally demanding and also requires expert knowledge to give accurate results. Accordingly, it is not presently used in high-throughput screening: fully automatized algorithms for setting up the calculations and determining convergence are lacking. In this paper, we develop such a method and, as a first application, use it to validate the accuracy of $G_0 W_0$ using the PBE starting point and the Godby-Needs plasmon-pole model ($G_0 W_0^{GN}$ @PBE) on a set of about 80 solids. The results of the automatic convergence study utilized provide valuable insights. Indeed, we find correlations between computational parameters that can be used to further improve the automatization of GW calculations. Moreover, we find that $G_0 W_0^{GN}$ @PBE shows a correlation between the PBE and the $G_0 W_0^{GN}$ @PBE gaps that is much stronger than that between GW and experimental gaps. However, the $G_0 W_0^{GN}$ @PBE gaps still describe the experimental gaps more accurately than a linear model based on the PBE gaps. With this paper, we hence show that GW can be made automatic and is more accurate than using an empirical correction of the PBE gap, but that, for accurate predictive results for a broad class of materials, an improved starting point or some type of self-consistency is necessary.

DOI: 10.1103/PhysRevB.96.155207

I. INTRODUCTION

Fifty years after its formal introduction by Hedin [1], and 30 years after the first applications to "real" solids [2–6], the GW method has become the standard approach for studying the electronic structure of solids. In various flavors, ranging from single shot G_0W_0 to fully self-consistent, it is included in many multipurpose first-principles codes [7–21].

Despite this long maturation period, performing even the simplest G_0W_0 calculation—not even considering the choice of starting point [22–31]—on a single solid is not a trivial exercise in any of the above-mentioned implementations. The complications appear at different levels.

(1) *GW* has a scaling of the computational complexity with system size (as represented by its number of electrons *N*) that is in common implementations at best $\mathcal{O}(N^4)$. Setting the computational parameters concerning memory and number of CPUs is hence much more involved than in typical density functional theory (DFT) calculations [with a scaling on $\mathcal{O}(N^3)$ or $\mathcal{O}(N^2 \ln N)$].

(2) *GW* shows a slower convergence with respect to the basis-set size as compared to DFT or Hartree-Fock. This worse convergence is linked to the need to accurately describe the cusp in the electron-electron pair correlation function, similar to the convergence of the random phase approximation (RPA) total energy methods [32,33]. Consequently, the results are

also more sensitive to the quality of pseudopotentials [34], projector-augmented wave data sets [35], resolution of the identity auxiliary basis sets [36], and the choice of local orbitals in full-potential linearized augmented-plane-wave approaches [37].

(3) GW calculations introduce additional, in some cases interlinked, computational parameters that need to be tested for convergence [38,39].

(4) In most implementations, a GW calculation is a threeto four-step process where the various input and output files need to be linked.

(5) Systematic validation studies using multiple codes and a wide variety of systems have recently been performed for solids at the DFT level [40] and for GW for molecules [36,41,42]. For GW calculations on solids, however, validation studies usually restrict to one code and only a limited amount of systems [35,43–45]. A systematic evaluation of the accuracy of the method is hence tedious at best.

Considering these difficulties and the lower general familiarity with the method, especially in the context of applications in the increasingly popular field of computational materials design and for high-throughput screening calculations, there is clearly a need for new automated approaches to perform GW calculations. Typical automatization schemes for DFT usually rely on overconverged safe computational settings or heuristic approaches [46–48]. For GW calculations especially, the worse scaling of the computational cost and memory do not allow for such an approach for all parameters. Hence, an automatic scheme is required to perform individual convergence studies for each compound.

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In this paper, we introduce such a framework for automatic GW calculations and apply it as a first example to about 80 solids establishing the accuracy and convergence properties of $G_0 W_0^{\text{GN}}$ @PBE: single shot perturbative $G_0 W_0$ starting from DFT results using the Perdew-Burke-Ernzerhof (PBE)[49] exchange correlations functional and using the Godby-Needs plasmon-pole model [50] (GNPPM) for the response function. It should be clear that choosing single shot G_0W_0 , i.e., neglecting any form of self-consistency, PBE as a starting point and GNPPM as an approximation to the full RPA response function, poses a limitation on the conclusions that can be drawn from the benchmark part of this paper. On top of that, the usage of the norm-conserving pseudopotentials, in contrast to full potential, all electron calculations introduce an approximation whose effects are still under debate. Removing these four uncertainties, however, is beyond the goal of the current paper. Here we establish the methodology needed for such future work, and take a first step for one of the simplest and most controllable flavors of GW.

We show that in our $G_0 W_0^{\text{GN}}$ @PBE results there are correlations between the values certain computational parameters need to take to reach convergence. This information can be used to further improve the automatization. From the validation study, we learn that there is a strong correlation between the error of the $G_0 W_0^{GN}$ @PBE gap (with respect to the experimental gap) and the experimental values. We also observe that this correlation is different for materials with or without transition metals. Moreover, we find that the correlation between the PBE gap and the $G_0 W_0^{\text{GN}}$ @PBE gap is stronger than the correlation between the $G_0 W_0^{GN}$ @PBE and the experimental gaps. In other words, the average error made by approximating the $G_0 W_0^{\text{GN}}$ @PBE gap from the PBE gap using the linear relation between them is smaller than the average error of $G_0 W_0^{\text{GN}}$ @PBE in reproducing the experimental gap. Finally, and most importantly, we show that $G_0 W_0^{GN}$ @PBE still outperforms a linear model trained on the experimental gaps, i.e., even in its simplest flavor, GW predicts the experimental gaps more accurately than an empirical correction based on the PBE gaps.

II. METHODOLOGY

In its most common approximation, which we adopt in this paper, the GW method provides corrections to the Kohn-Sham (KS) band structure via the linearized quasiparticle (QP) equation:

$$\epsilon_{n\mathbf{k}}^{\mathrm{QP}} = \epsilon_{n\mathbf{k}}^{\mathrm{KS}} + Z_{n\mathbf{k}} \langle n\mathbf{k} | \Sigma(\epsilon_{n\mathbf{k}}^{\mathrm{KS}}) - \mathcal{V}_{\mathrm{xc}} | n\mathbf{k} \rangle, \qquad (1)$$

where ϵ^{QP} and ϵ^{KS} are the QP and KS energies respectively, V_{xc} is the exchange correlation potential, $\Sigma(\epsilon)$ is the *GW* self-energy and

$$Z_{n\mathbf{k}} = \left(1 - \left\langle n\mathbf{k} \middle| \frac{\partial \Sigma(\epsilon)}{\partial \epsilon} \middle|_{\epsilon_{n\mathbf{k}}^{\mathrm{KS}}} \middle| n\mathbf{k} \right\rangle \right)^{-1}$$
(2)

is a renormalization factor originating from the linearization procedure. Adopting the usual decomposition of Σ in the energy-independent exchange and energy-dependent correlation part,

$$\Sigma(\epsilon) = \Sigma_{\rm x} + \Sigma_{\rm c}(\epsilon), \tag{3}$$

we have the following expressions for the diagonal matrix elements:

$$\langle \mathbf{k} | \Sigma_{\mathbf{x}} | n \mathbf{k} \rangle = -\frac{4\pi}{V} \sum_{\nu}^{\text{occ}} \sum_{\mathbf{q}}^{\text{BZ}} \sum_{\mathbf{G}}^{\Omega_{\mathbf{x}}} \frac{\left| M_{\mathbf{G}}^{n\nu}(\mathbf{k}, \mathbf{q}) \right|^2}{\left| \mathbf{q} + \mathbf{G} \right|^2}, \qquad (4)$$
$$\langle \mathbf{k} | \Sigma_{\mathbf{c}}(\omega) | n \mathbf{k} \rangle = \frac{i}{2\pi V} \sum_{\mathbf{q}}^{\text{BZ}} \sum_{\mathbf{G}_1 \mathbf{G}_2}^{\Omega_{\mathbf{c}}} \sum_{m=1}^{N_{\text{b}}} \left[M_{\mathbf{G}_1}^{mn}(\mathbf{k}, \mathbf{q}) \right]^{\dagger}$$
$$\times M_{\mathbf{G}_2}^{mn}(\mathbf{k}, \mathbf{q}) v_{\mathbf{G}_1 \mathbf{G}_2}(\mathbf{q}) J_{\mathbf{G}_1 \mathbf{G}_2}^{m\mathbf{k}-\mathbf{q}}(\mathbf{q}, \omega), \quad (5)$$

where v is the Coulomb potential in Fourier-space and $\Omega_{x/c}$ defines the sphere in G space for the exchange and correlation parts of Σ , respectively: $\mathbf{G} \in \Omega_{x/c}$ if $\frac{|\mathbf{q}+\mathbf{G}|^2}{2} \leq E_{c/x}$. In the remainder of this paper, these two cutoffs will be referred to as E_c and E_x for the correlation and exchange part cutoffs, respectively [51]. N_b denotes the number of KS orbitals used in constructing G_0 and W_0 , the same value is also used in constructing J. In the exact formulation, all occupied and infinitely many unoccupied single particle states would need to be included. In practice, the total number of available states is truncated by the finite number of basis functions for the single particle states. In constructing Σ , it is further constrained by N_b . Determining the values for E_c and N_b that lead to converged results is one of the most important steps in a GW convergence study. The matrix elements M are given by

$$M_{\mathbf{G}}^{b_1b_2}(\mathbf{k},\mathbf{q}) \equiv \left\langle \mathbf{k} - \mathbf{q}, b_1 | e^{-i(\mathbf{q} + \mathbf{G}) \cdot \mathbf{r}} | \mathbf{k}, b_2 \right\rangle$$
$$= \sum_{\mathbf{G}'} u_{\mathbf{k} - \mathbf{q}b_1}^{\dagger}(\mathbf{G}') u_{\mathbf{k}b_2}(\mathbf{G} + \mathbf{G}').$$
(6)

The matrix elements J in Eq. (5) originate from the screening and take a particular form depending on the approach used to describe the frequency dependency of W. In this paper, all calculations are performed using the Godby-Needs plasmon-pole approximation:

$$\Re \epsilon_{\mathbf{G}_{1}\mathbf{G}_{2}}^{-1}(\mathbf{q},\omega) = \delta_{\mathbf{G}_{1}\mathbf{G}_{2}} + \frac{\Omega_{\mathbf{G}_{1}\mathbf{G}_{2}}^{2}(\mathbf{q})}{\omega^{2} - \tilde{\omega}_{\mathbf{G}_{1}\mathbf{G}_{2}}^{2}(\mathbf{q})},$$
(7)

in which the parameters $\tilde{\omega}$ and Ω are derived to reproduce the *ab initio* inverse dielectric matrix ϵ^{-1} computed at the static limit and at an additional imaginary frequency point [6,11,12,50]. In terms of $\tilde{\omega}$ and Ω , the matrix elements of *J* are given by [52,53]

$$J_{\mathbf{G}_{1}\mathbf{G}_{2}}^{m\mathbf{k}-\mathbf{q}}(\mathbf{q},\omega) = \Omega_{\mathbf{G}_{1}\mathbf{G}_{2}}^{2}(\mathbf{q})\int \frac{e^{i\omega'\delta}\,\mathrm{d}\omega'}{(\omega+\omega'-\epsilon_{s}+i\eta\,\mathrm{sgn}(\epsilon_{s}-\mu))(\omega'^{2}-(\tilde{\omega}_{\mathbf{G}_{1}\mathbf{G}_{2}}(\mathbf{q})-i\eta)^{2})}.$$
(8)

All calculations presented in this paper are performed using the ABINIT software package [11], employing newly developed, optimized, norm-conserving, Vanderbilt pseudopotentials (ONCVPSP) [54]. Two projectors per angular momentum channel are used, in order to include semi core states while at the same time keeping a good description of the continuum states in the empty region, both of which are crucial for accurate GW calculations. In general, we observe a close agreement of the logarithmic derivatives up to energies of 300 eV. Deviations of the logarithmic derivative of the pseudized wave functions from that of the reference atomic all-electron calculation are an indication of possible ghost states. Manual inspection of the band structures of elemental solids up to 100 eV above the Fermi level did not reveal any ghost state in the ONCVPSP pseudopotentials used in this paper [54].

Our automatic GW workflow consists of the following steps:

(1) Convergence testing of the KS energies with respect to the basis set, i.e., the energy cutoff of the plane wave basis;

(2) Convergence testing of the QP energies with respect to the number of empty states, $N_{\rm b}$, and cutoff in the correlation part of Σ , $E_{\rm c}$, at a low density k-mesh;

(3) Testing of the convergence behavior of the QP energies at a high density k-mesh;

(4) Postprocessing of the results calculated at the highdensity k-mesh: constructing a scissor operator and band structure and storing it in a database.

The flow itself, its generation, execution, and postprocessing are programmed within the AbiPy python framework [12,55]. AbiPy is an open-source library for the analysis of the results produced by ABINIT based on the Python ecosystem powering the Materials project pymatgen [56]. The following sections describe and validate these steps in detail.

A. Convergence at the KS level

In the automatic flow, the energy cutoff for the wave functions is fixed at 44 Ha (1197 eV) and an automatic test up to 52 Ha (1415 eV) is conducted to ensure convergence. For none of the compounds, the convergence of the total KS eigenvalue energy range [57] indicated that a larger value was needed. The other, ground-state related, computational parameters are fixed in the flow to ensure convergence is reached within 0.05 eV on the full KS bandwidth.

B. Automatic convergence testing at the QP level

To optimize the performance of the entire flow, it is essential to decouple those parameters for which the convergence can be studied independently. In general, the energy cutoff of the correlation part of the self-energy Σ (E_c) and the number of empty states (N_b) are coupled and need to be considered simultaneously. The convergence properties of this pair and that of the k-mesh density are, however, decoupled. The value of E_x is fixed to the same 44 Ha as used for the KS calculation.

The convergence with respect to E_c and N_b is studied as a 2D problem at a low-density (Γ centered 2 × 2 × 2) k-point mesh. On a 4 × 4 ($E_c \times N_b$) grid of parameters, a full *GW* calculation is performed. First, for fixed E_c , the converged N_b



FIG. 1. The convergence of the gap at Γ as a function of the number of bands (N_b) and the cutoff used for the screening, and Σ (E_c) for Si (upper) and BN (lower).

values are determined by fitting an asymptotic function. To improve the stability of the fitting procedure, the algorithm fits multiple functions with only two parameters and selects the best fit [58]. Second, the final converged result is obtained by fitting the converged results at fixed E_c obtained in the previous step. This final value is used to determine for which pair (E_c , N_b) the result is within a predefined distance from the converged result. The pair (E_c , N_b) is chosen to minimize E_c to reduce the computational cost. If no (E_c , N_b) pair can be determined on the current grid, the grid is automatically extended to add more data points until the converged pair is found.

Figure 1 shows two examples of the convergence of the fundamental QP gap at Γ with respect to E_c and N_b for silicon and boron nitride. Both Si and BN show the typical coupled convergence behavior but the convergence rate is an order of magnitude larger for BN. This difference already



FIG. 2. The (E_c , N_b) convergence data summarized for seven k-point meshes, labelled by NK, corresponding to a NK x NK x NK grid. For each k-mesh, the average, maximum, and minimum deviation from the data at the largest density grid ($14 \times 14 \times 14$) is plotted. The data is shown for gold.

shows that employing heuristic rules to perform automatic GW calculations in several systems would be error-prone since the convergence rate is strongly system-dependent. Rules of thumb extracted from a restricted data set hence cannot be extrapolated to a large class of systems. The coupled behavior between E_c and N_b will be discussed in detail in Sec. III B.

The decoupling of the convergence with $E_{\rm c}$ and $N_{\rm b}$ and the density of the k-point mesh is shown in Fig. 2. Here, the $(E_{\rm c}, N_{\rm b})$ convergence data is summarized for seven k-point meshes for Au [59]. For this example, a metal was chosen for which the dependence on the k-mesh is larger than for the semiconductors and insulators used in this study. Because of this, no Drude contribution [60,61] is included in these specific calculations; the example is only used to show that the decoupling takes place even in the case of metallic occupation. For each k-mesh we plot the average, maximum, and minimum deviation from the results obtained with the highest density grid. The average shows how well the actual k-mesh gives converged results, while the difference between minimum and maximum deviation measures how well the shape of the convergence surface is described by a specific k-mesh. The small difference between the minimum and maximum with the $2 \times 2 \times 2$ mesh already shows that performing the convergence study using this coarse grid is valid.

The final computational step consists of a test to ensure that the convergence rate at the high-density k-point grid (HDG) is indeed similar to the one at the low-density grid (LDG). To this end, four full GW calculations are performed at a predefined HDG. From these four data sets, the derivative of the gap with respect to E_c and N_b is calculated by means of finite differences and compared to the corresponding values calculated at the LDG. In about 90% of the systems studied here, the E_c slopes at the HDG are actually lower than on the LDG. For the remaining systems, the E_c slopes are only marginally larger at the HDG, see Fig. 3. For N_b , the situation is less clear. However, the cases where the slopes are larger than 1 meV at the LDG do



FIG. 3. Relation between the E_c slopes using the Low Density Grid (LDG) and High Density Grid (HDG).

decrease in the HDG. For the smaller slopes, we observe a more scattered behavior, see Fig. 4. Hence, for all systems in this set, the converged parameters found on the LDG can be safely used on the HDG to obtain converged results. A full list of the numerical data is available in the Supplemental Material [62].

C. Postprocessing

After the final converged calculation is performed, the flow performs various postprocessing steps. The output of the final GW calculation is stored in a database as a queryable entry containing a link to a simultaneously stored NetCDF output file [12]. For the full QP spectrum, a bilinear energy-dependent extrapolation function is constructed. For this, an improved version of the polyfit method [63] is used. The first linear section passes through the valence band maximum



FIG. 4. Relation between the N_b slopes using the Low Density Grid (LDG) and High Density Grid (HDG).



Si Bandstructure

FIG. 5. The band structure of silicon computed at the KS level and with the energy-dependent extrapolation function described in the text.

(VBM) correction and optimally fits the corrections to the other occupied QP states. The second section passes through the conduction band minimum correction and optimally fits the corrections to the other unoccupied states [64]. The extrapolation function is then used to perform corrections to a separately calculated full band structure. Figure 5 shows the band structure of silicon obtained using this procedure, with the bands aligned at the VBM.

In the analysis of the complete data set, we look for relations between gaps at different levels of theory, converged input parameters, and other *a priori* known quantities. T-tests are used to establish whether the obtained results are statistically relevant. Regression obtained using ordinary least square (OLS) minimization and using robust linear models (RLM), using the Huber T norm [65], are both used to obtain the sought-after relations. In the latter, the deviations *d* weighted according to:

$$w(d) = 1; \ d \leq t,$$

$$w(d) = \frac{t}{d}; \ d > t,$$
(9)

using the default value of t = 1.345, are minimized instead of the bare deviations in the OLS. It is therefore less sensitive to outliers than OLS and, comparing the two approaches, gives an indication of the presence of real outliers.

III. RESULTS FOR $G_0 W_0^{\text{GN}}$ @PBE

As a first application, we use the above-described methodology on a set of solids used previously for evaluating various DFT functionals and the gap-prediction method Δ -sol [66], using the Godby-Needs plasmon-pole model as implemented in ABINIT [11,12,50]. Of this original set, those systems are included that are not metallic at the KS level and have a size that allowed a converged $G_0 W_0^{\text{GN}}$ @PBE given our current computational resources. In the final set, there are three elemental compounds and 75 binaries with stoechiometry occurrences: 51 XY, 13 XY₂, six XY₃, one XY₄, three X₂Y₃, and one X₂Y₅. Among the binaries, there are 12 III-V, 12 II-VI, 11 I-VII, and eight IV-VI compounds. Anions may also serve to characterise our set. There are 12 halides, 37 chalchogenides (including 12 oxides), and 20 pnictides (including four nitrides). The full results for all systems included in this study [67] are given in the Supplemental Material. A comparison will be made in this section to experimental gaps collected in Ref. [66]. To make the comparison in a proper way, there are, however, three different effects that should be taken into account.

First, the experimental values have all been measured at room temperature. As such, they have a tendency to underestimate the purely electronic gap, at 0 K without zero-point renormalization, by 0.1 to 0.4 eV [68-73]. For 31 of the systems presented in this section, we could obtain a 0 K extrapolated band gap using literature values [74]. For these, we find an average underestimation of 0.12 eV with a median of 0.09 eV. Second, the calculated results only include relativistic effects at the scalar relativistic level. For systems containing heavy elements, the inclusion of spin-orbit corrections can lead to reductions of the band gap by up to 0.5 eV [75,76]. Finally, in the convergence studies, the convergence criterion was set to 0.1 eV. Since, in almost all cases, the asymptotic value band gap is approached from below, in general, we underestimate the fully converged result by up to 0.1 eV. Taking the cumulative effect of these three effects into account leads to a window of 0–0.9 eV. This means that even the (scalar relativistically) exact electronic structure theory would overestimate the room temperature experimental gap by 0-0.9 eV. In comparisons made below, this interval is indicated by two blue lines, i.e., "perfect agreement with experiment" would lead to a data point between the two blue lines.

Besides these three sources of errors originating from approximations in the computational setup, the experimental results carry an uncertainty. For about half of the systems presented in this section we have found additional experimental room temperature (290–300 K) gaps in the literature [74]. Within these collected results, we find an average standard deviation of 0.16 eV, with a median of 0.08 eV.

The full data, details about all systems considered in this set, all numerical values, and full detail on the statistical analysis are available in the Supplemental Material.

A. Evaluation of the accuary of $G_0 W_0^{GN}$ @PBE

In Fig. 6, the calculated gaps are obtained from a calculation on high symmetry lines through the Brillouin zone. The $G_0 W_0^{\text{GN}}$ @PBE results are obtained from correcting the KS band structure using an energy-dependent extrapolation function, as described in Sec. II C.

In general, we observe the typical opening of the gap by $G_0 W_0^{GN}$ @PBE as observed in many previous studies for various flavors of GW [1,5,35,43,77,78]. However, we do observe a much larger spread of the GW error for this set of systems.

The comparison made in Fig. 6 is reevaluated in Fig. 7, subtracting the experimental values and adding various identifiers, identifying which compounds contain transition metals and the mass of the lightest and heaviest element present. Zeropoint renormalization and relativistic effects (not taken into



FIG. 6. Comparison of the KS and QP fundamental gaps evaluated from a band structure computed on a high-symmetry path through the Brillouin zone. The QP band structures are obtained from applying the extrapolation function to the KS band structure. The blue lines indicate the estimated interval in which exact agreement would be expected, taking spin-orbit coupling, finite temperature, and the level of convergence into account, see text.

account here) might be the origin of larger errors in systems with light and heavy elements, respectively. However, the identification of light and heavy element systems in Fig. 7 shows that this is not the dominating effect.



FIG. 7. The error of the GW gap with respect to the experimental value compared to the experimental value. The squares indicate compounds containing transition metals, the circles compound without transition metals. The size of the symbols reflects the smallest atomic number occurring in the compound; the color, the largest atomic number. The horizontal blue lines indicate the estimated interval in which exact agreement would be expected taking spin-orbit coupling, finite temperature, and the level of convergence into account.

TABLE I. Statistics on the deviation of the QP band gap from the experimental gap. Systems with a zero band gap in KS-PBE have been omitted. All gaps are in eV. Z^{max} and Z^{min} report the maximal and minimal atomic number present in the compound.

	$E_{\rm g}^{\rm EXP}$	$E_{\rm g}^{\rm KS}$	$E_{\rm g}^{\rm GW}$	$E_{\rm g}^{\rm GW}$ - $E_{\rm g}^{\rm EXP}$	Z^{\max}	Z^{\min}
count	75.00	75.00	75.00	75.00	75.00	75.00
mean	2.81	1.63	2.62	-0.18	46.40	21.20
std	2.38	1.69	2.26	0.63	23.01	13.93
min	0.13	0.02	0.44	-2.68	6.00	4.00
25%	1.15	0.48	1.09	-0.35	31.00	8.00
50%	2.10	1.00	1.93	-0.09	49.00	16.00
75%	3.17	1.93	2.94	0.15	55.00	32.50
max	10.59	7.67	10.10	1.30	83.00	55.00

Making the distinction on the presence of transition metal elements (elements for columns 3–12 of the periodic table), on the other hand, reveals a clear correlation of the $G_0 W_0^{\text{GN}}$ @PBE error with the experimental gap. However, the relation for an OLS and RLM using the Huber T norm with median absolute deviation scaling [65] is different between the two groups. For the transition metal compounds, we have $E_g^{\text{GW}}-E_g^{\text{EXP}}=0.42-0.37E_g^{\text{EXP}}(\text{R}^2=0.14)$ and $E_g^{\text{GW}}-E_g^{\text{EXP}}=0.29-0.28E_g^{\text{EXP}}$. For the nontransition-metal compounds we have $E_g^{\text{GW}}-E_g^{\text{EXP}}=0.09-0.07E_g^{\text{EXP}}(\text{R}^2=0.16)$ and $E_g^{\text{GW}}-E_g^{\text{EXP}}=0.13-0.07E_g^{\text{EXP}}$. The trends observed here agree well with Lany's observations on 3D-transition-metal oxides [79]. For the transition-metal containing-compounds, we observe a mean absolute deviation of 0.64 eV from the experimental gaps. For the compounds without transition metals this reduces to 0.38 eV.

Table I lists the statistical evaluation of the data presented in Fig. 7. We observe a 0.18 eV average underestimation of the GW results with respect to the room temperature experimental gaps. Adding 0.45 eV to, on average, correct for relativistic effects and finite temperature we get to a 0.63 eV underestimation.

The compounds with a $G_0 W_0^{\text{GN}}$ @PBE error larger than one standard deviation (0.64 eV) from the mean (-0.21 eV) are listed in Table II. At the top of the table, we find the compounds where GW is underestimating the most. We note that a significant fraction of these compounds contains copper. Besides these, we find GeS, SnO₂, and GaN strongly underestimating. Ge, Sn, and Ga occur in various other compounds in our set that do not stand out particularly. It could, however, be that, in these particular cases, the low-lying d levels are problematic for GW. In CaO and NaCl, the absolute value of the error is about 1; however, since the actual gaps are rather large, the relative errors are much smaller than in most compounds. In the lower part of Table I, we found compounds containing Te and Sn in which spin-orbit effects are expected to be strong. V₂O₅ will be discussed below.

1. Specific systems

 V_2O_5 . The largest overestimation in our data set is observed in V_2O_5 , in which G_0W_0 gives a gap of 3.50 eV that significantly exceeds the experimental value by 1.30 eV. This material has been studied by means of *GW* calculations

TABLE II. Compounds for which the $G_0 W_0^{GN}$ @PBE error with respect to the room temperature experimental gap is more than one standard deviation (0.64eV) from the mean deviation (-0.21eV). Type indicates the classification as transition metal containing (2) or not (1). All gaps are in eV. Z^{max} and Z^{min} report the maximal and minimal atomic number present in the compound.

System	Туре	$E_{\rm g}^{\rm EXP}$	$E_{\rm g}^{\rm KS}$	$E_{\rm g}^{\rm GW}$	$E_{\rm g}^{\rm GW}$ - $E_{\rm g}^{\rm EXP}$	Z^{\max}	Z^{\min}
CuCl	2	3.30	0.50	0.62	-2.68	29	17
CuBr	2	2.91	0.43	0.64	-2.27	35	29
GeS	1	1.65	0.00	0.00	-1.65	32	16
SnO_2	1	3.73	1.21	2.34	-1.39	50	8
RuS_2	2	1.80	0.69	0.47	-1.33	44	16
CuI	2	2.95	1.10	1.79	-1.16	53	29
RbAu	2	2.50	0.25	1.37	-1.13	79	37
GaN	1	3.44	1.94	2.31	-1.13	31	7
CaO	1	6.93	3.66	5.86	-1.07	20	8
NaCl	1	8.97	5.11	8.01	0.96	17	11
NiP ₂	2	0.73	0.24	-0.21	-0.94	28	15
Bi ₂ Te ₃	1	0.13	0.35	0.58	0.45	83	52
TlSe	1	0.73	0.41	1.21	0.48	81	34
PtS	2	0.80	0.45	1.30	0.50	78	16
BAs	1	1.46	1.20	2.11	0.65	33	5
BeTe	1	2.80	1.98	3.45	0.65	52	4
SnSe ₂	1	0.97	0.68	1.68	0.71	50	34
TePb	1	0.31	0.74	1.05	0.74	82	52
V_2O_5	2	2.20	1.81	3.50	1.30	23	8

previously: Lany [79], using $evGW_0@GGA+U$ [80], found a gap of 4.69 eV and Bhandari *et al.* [81], using qs*GW* [82], found a gap of 4.0 eV. Both (partial) self-consistent approaches are known to enlarge the gap as compared to G_0W_0 . Both results, in this sense, agree with our 3.50 eV G_0W_0 gap. Bhandari *et al.* hypothesized the discrepancy with respect to the experimental value to be caused mainly in terms of the effects of lattice polarization [81].

ZnO. Wurtzite ZnO is a well-known problematic system for *GW*. [34,38,83–87]. The physical model, i.e. the approximation used for the response function (plasmon pole or full frequency), has a large influence on the band gap [38], but also the convergence with respect to the number of unoccupied states is much slower in ZnO [38,86] than in other solids such as silicon. Indeed, our calculations confirm this behavior. Within the computational resources defined in our algorithm, the procedure does not find a set of converged parameters before the calculations become unfeasible, given the computing capabilities available to us [88].

2. The correlation of KS-PBE and $G_0 W_0^{GN}$ @PBE gaps

The relation between the KS-PBE and $G_0 W_0^{\text{GN}}$ @PBE fundamental gaps (on the regular k-point grid) is investigated in Fig. 8. In contrast to the comparison of the $G_0 W_0^{\text{GN}}$ @PBE gaps with the experimental values, the data do not show a significant difference between the two groups of materials, i.e., those containing transition metals and those without transition metals. The correlation between KS-PBE and $G_0 W_0^{\text{GN}}$ @PBE gaps is, however, stronger than any other observed in the data. Moreover, in performing the linear regression analysis we find



FIG. 8. Comparison of KS-PBE and $G_0 W_0^{\text{GN}}$ @PBE gaps. An ordinary least squares (OLS) linear regression leads to an $E_g^{\text{GW}} = 0.51 + 1.32 E_g^{\text{KS}} (\text{R}^2 = 0.97)$ relation between the two sets with a correlation of 0.99. A robust linear model (RLM) leads to only slightly different results: $E_g^{\text{GW}} = 0.53 + 1.31 E_g^{\text{KS}}$.

remarkably few outliers; the OLS and RLM find the same parameters of the linear relation.

The relation found between the KS-PBE and $G_0 W_0^{\text{GN}}$ @PBE gaps can be used to estimate the QP gap directly from the KS-PBE gap. The mean absolute deviation of the estimated $G_0 W_0^{\text{GN}}$ @PBE gap from the actual values are 0.32, 0.37, and 0.29 eV for the whole set, transition metal-containing compounds, and compounds without transition metals, respectively.

Beyond the correlations between the experimental, KS-PBE, and $G_0 W_0^{\text{GN}}$ @PBE gaps, the next strongest correlations are observed between the gaps and the ratio of the electronegativities of the elements present in the compounds (see "All correlations in the dataset" in the Supplemental Material). The experimental gaps show the strongest correlation with the electronegativity ratio: 0.60.

3. $G_0 W_0^{GN}$ @PBE vs an empirical correction to KS@PBE

An important question in addressing the accuracy of $G_0 W_0^{\text{GN}}$ @PBE is whether it performs better or worse than a linear model based on the KS gaps fitted to the experimental values, as proposed by Setyawan *et al.* [89]. Figure 9 shows this comparison. The performance of KS-PBE, $G_0 W_0^{\text{GN}}$ @PBE and the linear model is reported in the inset.

The linear regression model is tested using fivefold cross validation. The cross-validation score for the model is 0.88 (full details are provided in the Supplemental Material). Both the standard deviation (SD) and the mean absolute error (MAE) of the GW results are significantly smaller than those of the linear model. The p-value for a paired t-test [90] between the GW and LM errors is 0.0006, indicating a significant difference between the two sets of results. The MAE and SD stay almost the same. We hence conclude that GW, even the simplest form of the GW method (perturbative single-



FIG. 9. Comparison of KS and GW gaps, and gaps obtained from a fivefold cross-validated linear model of the KS gap trained on the experimental gaps. The statistical evaluation provides the mean absolute error (MAE), mean error (ME), and standard deviation (SD).

shot correction based on the plasmon-pole model), already significantly outperforms a single parameter empirical model fitted to the experimental results. If the linear model is trained using the GW gaps, the model is only better for the mean error (ME). Only when we train a model including both the type, electronegativity ratio, the nuclear mass ratio, and the KS gap do we reach a model with similar accuracy as GW: 0.43, -0.03, and 0.59 eV for the MAE, ME, and SD respectively.

4. Comparison of $G_0 W_0^{GN}$ @PBE to Δ -sol and HSE06.

A computationally efficient method to significantly improve KS gaps was proposed by Chan and Ceder in 2010 under the name of Δ -sol [66]. It is closely related to the Δ -SCF method for finite systems where the HOMO-LUMO gap is calculated explicitly as the change in total energy upon changing the number of electrons in the system. In the Δ -sol method, one electron is removed per screening volume. How many electrons effectively reside in a screening volume, N^* , depends on the exchange-correlation functional and the type of system. In Ref. [66], the values of N^* are determined for systems containing sp and spd valence electrons for three functionals. In addition, the N^* are specifically geared to be predictive in a range of gaps of between 0.5 and 4 eV. Using these parameters, mean absolute errors of 0.31 eV for a set of typical semiconductors, and 0.26 eV for a larger set of transition-metal-containing compounds are found. The errors for gaps calculated using HSE06[91-94] (0.26 and 0.41 eV) are also reported. The $G_0 W_0^{GN}$ @PBE gaps reported in this paper hence have larger mean absolute errors than the gaps obtained using Δ -sol and HSE06. The 0.44 eV MAE of $G_0 W_0^{GN}$ @PBE, however, is achieved for systems with gaps up to 10 eV. If we only consider the 0.5-4.0 eV gap range for the nontransition metal compounds in our set, the $G_0 W_0^{\text{GN}}$ @PBE MAE drops to 0.32 eV, and the MAE for the compounds containing transition metals in this range increases to 0.61 eV. Summarizing, we conclude that $G_0 W_0^{GN}$ @PBE has similar

TABLE III. Statistics on the convergence parameters.

	$E_{ m c}$	$E_{\rm c}^{\rm interpol}$	$N_{ m b}$	E^{\max}
count	78.00	78.00	78.00	78.00
mean	244.28	150.26	282.82	117.79
std	60.45	78.29	278.36	72.43
min	93.11	93.87	35.00	33.09
25%	212.05	105.33	105.00	64.43
50%	216.66	108.38	215.00	80.71
75%	315.46	178.46	342.50	155.65
max	432.87	434.53	1,505.00	338.67

accuracy in predicting band gaps as HSE06 and Δ -sol for *sp* compounds. For compounds containing transition metals, the HSE06 and Δ -sol methods have a better accuracy.

B. Convergence parameters

The significant number of GW results obtained with our automatically converged studies allow us to investigate possible relationships among the values needed to reach convergence. A statistical summary of the converged parameters is given in Table III. The median is for all observables clearly larger than the mean value, indicating an asymmetric distribution in all cases.

Calculating the correlation coefficients between all quantities in our data set reveals two significant correlations involving the convergence parameters. The first is the correlation between the cutoff for the expansion of the screening and the self-energy (E_c), and the energy of the highest band (E^{max}). It is investigated in Fig. 10. Since the grid of E_c values actually used is relatively coarse, the statistics are performed on an interpolated E_c , $E_c^{interpol}$, which estimates the actual value for which the convergence criterium would be met. In contrast, E_c is the first value on the grid for which it is met. The correlation between the two parameters is 0.53. There is a clear difference between the OLS and the RLM fit, with the RLM looking more appropriate. Although the fit is only barely significant, the ob-



FIG. 10. Comparison of the energy of the highest band and the energy cutoff for Σ needed to reach convergence.



FIG. 11. Comparison of the energy of the highest band included in the construction of Σ needed to reach convergence and the KS gap.

tained results can be used to simplify convergence studies for new materials. Instead of treating the cumbersome 2D convergence problem, one could perform a convergence study on both at the same time by keeping the ratio between the two fixed.

A relation between the maximum number of bands and the cutoff energy for correlation has been observed for individual systems before [38,39], although, to our knowledge, it has not been studied in a systematic way yet. It may be rationalized as follows. The expression for the RPA dielectric matrix reveals that the **G** dependence of the screening is defined by the oscillator matrix elements [Eq. (6)] that are given by a convolution in **G** space between the periodic parts of the Bloch states associated to one occupied and one empty state. As a consequence, the convergence of the screening matrix elements with large **G**₁ or **G**₂ is expected to be governed by the inclusion of high-energy (free-electronlike) KS states in the sum over empty states.

In 62% of the systems, the $\Gamma - \Gamma$ gap converges towards a larger gap in both parameters, as is the case for BN (Fig. 1). In the remaining systems, we observe opposite convergence directions for E_c and N_b , where E_c is slightly more often converging towards a larger gap. In about 23% of the systems, some form of nonmonotonous behaviour is observed. In the cases where the convergence in N_n is nonmonotonous, this happens at a low number of bands and is rather sharp, as is seen for Si in Fig. 1. The exact nature of this nonmonotonous behaviour in $N_{\rm b}$ seems to depend on particular features in the band structure that are hard to predict a priori. In the cases where it occurs in E_c , it occurs at high E_c values and is usually not very strong, creating just a slight maximum value for the gap. There are even cases (e.g., WS₂) where the gap converges with respect to E_c , upwards for low values of N_b and downwards for high values of $N_{\rm b}$.

The correlation between the KS gap and the energy of the highest band (see Fig. 11) is the second significant correlation. But with a correlation coefficient of only 0.35, it clearly looks less promising. The large fraction of outliers and large spread make it difficult to extract any heuristic rule.

IV. CONCLUSION

We have presented a methodology to perform GW calculations in an automatic manner requiring as little human intervention as possible. The most important aspects are the coupled 2D convergence study on the number of empty states N_b and the energy cutoff of the screening and self-energy E_c , the decoupling of this convergence study, and the density of the k-point mesh. We apply this methodology to ~80 solids establishing the accuracy and convergence properties of $G_0W^{GN}0@PBE$: single shot perturbative G_0W_0 using the KS-PBE [95] as a starting point within the Godby-Needs plasmon-pole model [50].

We confirm that the convergence behavior of N_b and E_c are connected. For a large part of the systems in our study, we observe that studying the convergence of one at low value of the other leads to underconverged parameters. We observe a small positive correlation between E_c and N_b . It indicates that a convergence study could be done on a single parameter keeping the relation between the two fixed. This, however, does not hold in general. A single-parameter convergence study may hence lead to overconverged parameters, leading to unnecessarily high computational costs.

The correlation between the $G_0 W_0^{\text{GN}}$ @PBE error (the difference between the $G_0 W_0^{\text{GN}}$ @PBE and experimental gap) and the experimental gap is clear if the compounds are separated, based on whether or not transition metals are present in the compound. For the transition metal compounds, we find the relation $E_g^{\text{GW}}-E_g^{\text{EXP}} = 0.29 - 0.28E_g^{\text{EXP}}$. For the nontransition-metal compounds, we observe a smaller error and a less strong dependency $E_g^{\text{GW}}-E_g^{\text{EXP}} = 0.13 - 0.07E_g^{\text{EXP}}$.

The correlation between the QP and KS gaps is very strong, $E_g^{GW} = 0.51 + 1.32 E_g^{KS}$ (R² = 0.97), and contains almost no outliers. Moreover, it does not show a separation between the transition metal and nontransition-metal-containing compounds. The error of $G_0 W_0^{GN}$ @PBE with respect to the experiment (MAE of 0.64 and 0.38 eV, transition-metal and nontransition-metal compounds respectively) is actually larger than the error of approximating the QP gap via its relation with the KS gap (MAE of 0.37 eV and 0.29 eV, transition-metal and nontransition-metal compounds respectively).

Finally, we build a linear model to predict the experimental gaps from the calculated KS gaps and compare the predictive power of this model to that of $G_0 W_0^{\text{GN}}$ @PBE. We find that both the mean absolute error and the standard deviation of the *GW* results are smaller than those of the linear model.

ACKNOWLEDGMENTS

The present research benefited from computational resources made available on the Tier-1 supercomputer of the Fédération Wallonie-Bruxelles, infrastructure funded by the Walloon Region under Grant No. 1117545. Financial support for M.G. was provided from F.R.S.-FNRS through PDR Grant No. T.0238.13 (AIXPHO). This work was supported by the Communauté française de Belgique through the BATTAB project (ARC 14/19-057). The work was supported by the European Union Marie Curie Career Integration (CIG) grant HTforTCOs PCIG11-GA-2012-321988.

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